# Evidence of a large-scale mechanosensing mechanism for cellular adaptation to substrate stiffness

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Cell migration plays a major role in many fundamental biological processes, such as morphogenesis, tumor metastasis, and wound healing. As they anchor and pull on their surroundings, adhering cells actively probe the stiffness of their environment. Current understanding is that traction forces exerted by cells arise mainly at mechanotransduction sites, called focal adhesions, whose size seems to be correlated to the force exerted by cells on their underlying substrate, at least during their initial stages. In fact, our data show by direct measurements that the buildup of traction forces is faster for larger substrate stiffness, and that the stress measured at adhesion sites depends on substrate rigidity. Our results, backed by a phenomenological model based on active gel theory, suggest that rigidity-sensing is mediated by a large-scale mechanism originating in the cytoskeleton instead of a local one. We show that large-scale mechanosensing leads to an adaptative response of cell migration to stiffness gradients. In response to a step boundary in rigidity, we observe not only that cells migrate preferentially toward stiffer substrates, but also that this response is optimal in a narrow range of rigidities. Taken together, these findings lead to unique insights into the regulation of cell response to external mechanical cues and provide evidence for a cytoskeleton-based rigidity-sensing mechanism.

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ell migration is not only sensitive to the biochemical composition of the environment, but also to its mechanical properties. Cells directly probe the physical properties of their environment, such as substrate stiffness, by pulling on it. Increasing evidence show that matrix or tissue elasticity has a key role in regulating numerous cell functions, such as adhesion (1), migration (2) and differentiation (3). Such functions are affected by cell-generated actomyosin forces that depend on substrate stiffness through a feedback mechanism (4). The sensitivity of cells to mechanical properties of the extracellular matrix (ECM) arises from the mechanosensitive nature of cell adhesion. Numerous plausible candidates for the transduction of mechanochemical signals have been tested (5). Among them, focal adhesions (FAs) appear to be the most prominent, as shown by the reported correlation between their area and sustained force exhibiting a constant stress (6-9). This mechanosensitivity is usually accounted for by a generic local mechanism in which a force applied to an FA induces an elastic deformation of the contact that triggers conformational and organizational changes of some of its constitutive proteins, which in turn can enhance binding with new proteins enabling growth of the contact (10-12). However, how this local mechanosensitivity can result in the ability of cells to sense and respond to the rigidity of their surroundings (2, 3, 13, 14) at a large scale remains largely unknown (5, 15). Much conflicting evidence has emerged from a variety of studies, leading to important questions. Not only FAs but also the opening of calcium ion channels (2, 16, 17) could participate in the build-up of cell tension in response to mechanical cues. Moreover, recent modeling (18) as well as indirect observations (19, 20) suggest that the contractile actomyosin apparatus can act as a global rigidity sensor (21). From a physical point of view, the deformation of the surrounding matrix in response to cell contractility is poorly understood; plausible mechanisms of cell mechanosensitivity imply that the regulation could be either mediated by the stress exerted by cells, or by the strain in the ECM (7, 22–24). These intriguing questions are currently intensively debated, because the detailed mechanisms of force transduction in response to ECM might explain the observed discrepancies in adhesion (1, 25), migration (2, 26), and differentiation (3, 27) of cells in environments of different rigidities and over different time scales.

#### **Results and Discussion**

Dynamics of Focal Adhesions and Traction Force Measurements on Substrates of Various Stiffnesses. Here we report real-time measurements of explicit correlations between traction forces and the formation of FAs as a function of substrate stiffness. We used microforce sensor arrays (µFSA) (28) together with epifluorescence microscopy of REF52 fibroblast cells expressing a fluorescently tagged FA protein (YFP-paxillin) (Fig. 1 A and B). We seeded REF52 cells stably expressing YFP-paxillin on  $\mu$ FSA with pillars of various diameters and heights coated with fibronectin (13, 29, 30). These different-shaped pillars result in substrates with various spring constants, k, from 3 to 80 nN/ $\mu$ m (Fig. 1C, Movie S1) without altering the molecular scale properties of its surface. After allowing cells to adhere on the substrate for at least approximately 5 h, cells developed traction forces oriented toward their center that caused a deflection of the micropillars (Fig. 1B). We simultaneously analyzed the dynamics of the traction forces and FA patches over time. As a control experiment, we compared the size distributions of FAs on µFSA of different rigidities to those on a flat polydimethylsiloxane (PDMS) surface (Fig. S1). As expected from previous studies (23, 25, 31), these distributions were skewed toward larger values for substrates of increased stiffness. The topography of 2-µm pillar substrates did not significantly affect the size of FAs, as shown by the similar size distributions of FAs on µFAS and flat PDMS substrates (Fig. S1).

We analyzed traction forces in the proximity of the cell edge where the highest forces as well as the largest FAs were observed. In most cases, force generation and FA assembly over time were

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**Fig. 1.** Cell adhesion and traction forces developed by REF52 fibroblasts expressing YFP-paxillin on micropillar substrates. (*A*) Scanning electron micrograph image of a typical REF52 cell on a micropillar substrate. (Scale bar, 15  $\mu$ m.) (*B*) Epifluorescent image of a single cell deforming the micropillar substrate (here of spring constant  $k = 34 \text{ nN}/\mu\text{m}$ ). Micropillars are labeled by Cy3-fibronectin (red), and YFP-paxillin-rich patches are in green. (Scale bar, 15  $\mu$ m.) (*C*) Sequential images of the insert area of *B* showing the dynamics of FA growth and micropillar displacements. (Scale bar, 10  $\mu$ m.) (*D*) Schematic representation of the experimental setup showing the formation of FAs on the top of a PDMS micropillar. (*E*) Typical example of the formation of an FA area (red) and the buildup of force (blue) as a function of time (on a substrate of 34 nN/ $\mu$ m).

tightly correlated (Fig. 1 *D* and *E*): An increase of both forces and FA sizes followed by a saturation phase occurred within a timescale of minutes (see an example on a substrate of 34 nN/µm, Fig. 1*E*). To characterize the dependence of such dynamics on substrate stiffness, we performed the measurements over a broad range of rigidities (from around 4 to 80 nN/µm). For each time point, we averaged the force, *F*, over different FAs present at the cell edge of different cells ( $n \ge 5$ ). Average forces,  $\langle F_k \rangle$ , exerted at paxillin sites, grew more rapidly and reached higher saturation values on substrates with higher rigidities (Fig. 2*A*). In contrast to the force, the dependency of FA growth (area) was less obvious although the saturation size depended on the stiffness of the pillars (Fig. 2*B*).

In addition, the saturation force  $F_p$  was found to be proportional to the stiffness of the substrate within our range of rigidities (Fig. 2C). This result implied that saturation forces correspond to a constant deformation of the substrate of around  $0.84 \pm 0.03 \mu m$ , in agreement with previous studies on other cell types (13, 23). Interestingly, the initial rate of force increase over time, dF/dt, also varied linearly with substrate rigidity, giving a speed of micropillar displacement equal to 1.3 nm/s independent of the stiffness. All together, these data suggest the existence of a mechanosensitivity mechanism that is regulated by substrate deformation.

Relationship Between Focal Adhesion Area, Force, and Stress. To further assess the nature of force transmission, we investigated the relationship between force and focal adhesion area as a function of substrate stiffness. The relationship between force and area for all focal adhesions at all time points is plotted in Fig. 3A for substrate rigidities from 4 to 80 nN/µm (each point represents an average over at least 15 FAs of approximately equal area). This range of stiffnesses corresponds to an equivalent Young's modulus,  $E_{\rm eff}$ , of the micropillar substrate varying from 2.8 up to 60 kPa (13). We clearly obtained a linear dependence between traction force and FA area at a given rigidity. This dependence, however, is not the same for each stiffness; surprisingly, it appears that different force values could be reached for the same FA area. For instance, an FA area of approximately  $2 \,\mu\text{m}^2$  leads to forces that vary from 3 up to 42 nN within a stiffness range from 4.7 up to 80 nN/µm (Fig. 3A). The fact that FAs of similar areas can sustain different forces, illustrated in Fig. 3 C and D, is compatible with the results of ref. 24 where a variability of the stress between FAs at different positions was found, but only for a given substrate rigidity. These results overall contrast with the commonly accepted understanding of FA mechanosensitivity by a local force-dependent mechanism, which usually assumes a constant stress (6–8). Interestingly, the extrapolation of this linear dependence to zero gave rise to nonzero forces. This



**Fig. 2.** Traction forces and focal adhesion dynamics. (A) Average traction forces as a function of time for different substrate stiffnesses. Data were pooled from  $n \ge 5$  different cells and  $n \ge 15$  different pillars. Each curve exhibits an initial regime of fast force increase followed by a saturation regime. (B) Average focal adhesion area for different stiffnesses as a function of time. (C) Saturation force as a function of substrate stiffness. A linear relationship was observed between the maximal force and the substrate stiffness up to 80 nN/µm, showing that cells maintained a constant deformation. The fit corresponds to the theoretical model based on the active stress exerted by cytoskeleton remodeling (see *SI Text*). (D) Rate of increase of force with time, dF/dt, as a function of stiffness; dF/dt was obtained by fitting the slope of the initial linear regime of the force curve as a function of time. (A). The rate of force with time, dF/dt, was proportional to the stiffness in the range of stiffnesses used in our study (up to 80 nN/µm).



Fig. 3. Relationship between stress and focal adhesion area. (A) Correlation between forces for different stiffnesses of the pillars and FA areas obtained by averaging data at all time points. Different forces are reached for the same size of FA on substrates of different stiffnesses. (B) Relationship between stress ( $\sigma = F/S_{FA}$ ) and substrate stiffness. An initial regime of fast increase of stress was observed for stiffness values below 34  $nN/\mu m$  followed by a slow increase up to 80 nN/ $\mu$ m. The gray circle corresponds to the stress value obtained on soft continuous PDMS substrate with a Young's modulus of 12 kPa by Balaban et al. (7). The equivalent rigidity of the micropillar substrate,  $E_{eff}$ , was computed according to our modeling (13). The stress ( $\sigma$ )-stiffness (k) relationship is fitted according to our model (see SI Text). (C and D) Typical examples of an REF52 cell spread on a soft substrate (12 nN/µm) and on a stiffer substrate (54 nN/µm). Micropillars are in red and paxillin stainings in green. (Scale bar, 8  $\mu$ m.) Focal adhesions of similar sizes ( $\approx$ 2.8  $\mu$ m<sup>2</sup>) on both conditions give rise to different forces of around 41 nN on the stiff μFSA and 16 nN on the soft one (see SI Text). (Force bar, 10 nN.)

observation could be explained by invoking a second regime of tension for FAs smaller than  $0.3 \ \mu\text{m}^2$ . This residual area of  $0.3 \ \mu\text{m}^2$  is similar to the typical size of focal complexes, which are able to sustain large traction forces that do not correlate with adhesion size (6, 32).

We then computed the stress,  $\sigma$ , defined as the force divided by the area of paxillin-rich patches. The stress corresponds to the slopes of the different curves representing the force as a function of FA area in Fig. 3A. By plotting the stress as a function of the rigidity (Fig. 3B), we discovered that the stress did not remain constant, but instead was an increasing function of the stiffness. Such an influence of substrate rigidity on the active cell response is consistent with original reports by Pelham and Wang (31). More precisely, we observed an initial regime corresponding to a fast increase of stress up to a value of 7 nN/ $\mu$ m<sup>2</sup> (1 nN/ $\mu$ m<sup>2</sup> = 1 kPa) below a stiffness of 34 nN/µm and was followed by a saturation phase for higher stiffnesses. In a previous study, Balaban et al. (7) concluded that the mechanical regulation of cell adhesion was governed by a constant stress, but their experiments were performed at a single stiffness. We indeed found that their result was well correlated to our stress-stiffness relationship according to the substrate stiffness used in their study (Fig. 3B). Taken together, our data indicate that the mechanosensitivity effect that we report in this study is not due to the previously reported mechanism mediated by FAs (6, 7, 33, 34), but requires a larger-scale mechanism.

Modeling of a Large-Scale Mechanosensing Process. The mechanism usually invoked to account for a local mechanosensitivity is that a force applied to an FA induces an elastic deformation of the contact at the molecular scale that triggers conformational and

organizational changes (such as unfolding) of some of its constitutive proteins, which in turn can enhance binding of new proteins enabling growth of the contact (11). Such a mechanism should therefore depend on the local stiffness of the adhesive substrate at the submicrometer scale (33) which corresponds to the surface of the top of the micropillar in our case, characterized by the elasticity of the PDMS, i.e., its Young's modulus,  $E_s$  (SI Text for more details on our phenomenological model). Consequently, if the mechanical feedback was mediated by the FA itself, then the stress applied by FAs would depend on  $E_s$ only and not on k, which can be varied independently of  $E_s$  in our set-up through the geometrical parameters of the pillars (SI Text). However, we show here that the stress,  $\sigma$ , depends on the spring constant of the pillars k (for the same  $E_s$ ). Such a dependency provides direct evidence that an extra feedback involving structures other than FAs must be involved. We suggest that this feedback could be mediated on a larger scale by the cytoskeleton, and in particular by the stress fibers pulling on FAs. We argue that a deformation of the cytoskeleton, such as a shortening, x, of the length of a stress fiber, has an impact on the active stress exerted by the cytoskeleton. Generally, this mechanical activation can be justified within the framework of the theory of viscoelastic active gels (35, 36). This phenomenological theory states that in nematic (or polar) media driven out of equilibrium such as the cytoskeleton, there exists an active stress  $\sigma_{ii}^a$  (here due to actin/myosin contractility) proportional to the nematic order parameter tensor  $Q_{ij} = \langle n_i n_j \rangle \delta_{ij} / 3$  where the vector *n* denotes the local orientation of actin filaments. The coupling of  $Q_{ij}$  to the strain  $U_{ij}$ , which is standard in nematic gels (37), then implies that the stress  $\sigma_{ii}^a$  depends on  $U_{ij}$  and therefore on the deformation x. In other words, a deformation x induces a strain  $U_{ii}$  in the cytoskeleton and therefore a reorganization of the orientation  $Q_{ij}$ of the filaments, which in turn yields an active stress.

For this coupling of  $Q_{ij}$  to the strain  $U_{ij}$  to be valid, the relaxation time of  $Q_{ij}$  has to be long, implying an elastic like rheology of the cytoskeleton. In fact, the rheology of the cytoskeleton is very complex and fluidization has been observed, but only in response to a transient stretch (38, 39). Our case is more like that of a sustained constraint, in which strain stiffening and long relaxation times have been reported (40–42), justifying our assumption.

Actin Organization Depends on Substrate Stiffness. To evaluate  $Q_{ii}$ for substrates of various stiffnesses, we performed experiments to image actin stress fibers on fixed cells spread on soft and stiff  $\mu$ FSAs. The polarization of the cell was then quantified by the scalar order parameter S defined by  $S = Q_{n0n0}$ , where  $n_0$  is the average direction of stress fibers. We found that in a soft case ( $\approx 4$  nN/µm), the stress fibers were fully disordered, leading to  $\langle S \rangle = 0.19 \pm 0.13$ , whereas they appeared fully polarized in a stiff case ( $\approx 100 \text{ nN/}\mu\text{m}$ ) leading to  $\langle S \rangle = 0.80 \pm 0.09 \text{ (SI Text)}$ . At intermediate stiffness ( $\approx 40 \text{ nN}/\mu\text{m}$ ), we obtained  $\langle S \rangle =$  $0.56 \pm 0.21$ . This result shows a relationship between force generation by cells and actin stress fiber alignment, yielding a semiquantitative support of the coupling between  $Q_{ij}$  and  $U_{ij}$  (Fig. 4). This active stress is a deformation-dependent (and therefore stiffness-dependent) contribution, enhancing the stress exerted by the FA. The existence of such coupling shows that actomyosin forces are sufficient to explain the cellular response to matrix rigidity according to a stress-dependent relationship to substrate stiffness, as recently shown for the response of stem cells to matrix stiffness (43). By assuming a linear coupling between  $Q_{ii}$  and  $U_{ij}$ , the model could be made explicit for the case of a single stress fiber, for which the stress,  $\sigma_{ij}^a$  at each extremity was given by the deformation of the corresponding pillar and the adhesion area,  $S_{\rm FA}$ . Based on this assumption, we made explicit the dependence of the stress, therefore the force, on the pillar rigidity and showed that this theoretical prediction, compatible with ref. 43 (SI Text), was in good agreement with our experimental data

Trichet et al



**Fig. 4.** Coupling between the organization of stress fibers and the stiffness of the substrate. (A-C) Schematic representations and actin organization on substrates with various rigidities. Typical examples of actin staining and stress fibers organization (green) on a soft substrate ( $\sim 4 \text{ nN}/\mu\text{m}$ ) (A), on a substrate with an intermediate stiffness ( $\sim 40 \text{ nN}/\mu\text{m}$ ) (B), and on a stiffer substrate ( $\sim 100 \text{ nN}/\mu\text{m}$ ) (C). The scalar order parameter, S, is given here by  $S = (\cos 2(\theta-\theta_0))$ , where  $\theta$  denotes the angle of a stress fiber with a reference axis and  $\theta_0$  its average. Cells on stiffer substrate exhibit a more pronounced polarization of the actin stress fibers leading to an increase of S with the rigidity:  $S \approx 0.1$ , 0.47, and 0.95, respectively. (Scale bar, 15  $\mu\text{m.}$ ) (D) Average order parameter for cells plated on substrate sof three different stiffnesses as a function of the pillar spring constant and the effective rigidity,  $E_{\text{eff}}$ , of the substrate. Data were pooled from  $n \ge 15$  cells.

(Fig. 2C, Fig. 3B). Additionally, this large-scale mechanosensitive regulation by the contractility of actin-myosin fibers could also explain the linear relationship that we obtained between the saturation force and stiffness (Fig. 2C): The constant deformation of around 840 nm could be attributed to the simultaneous shortening of several micron-sized sarcomeric substructures within actomyosin stress fibers that, according to previous studies (44, 45), can sustain contractions of 10–25%.

Substrate Stiffness Governs the Directionality of Cell Movements. Such a large-scale mechanism driven by cell cytoskeleton polarization could also provide a plausible explanation for cell durotaxis. The combination of our observations-a faster force increase and higher saturated force values on stiffer substrates and an increasing stress with respect to substrate stiffness-led us naturally to the following prediction: polarized cells coming from a softer substrate will suddenly exert large traction forces as they touch the stiff side, and thus rotate to migrate perpendicularly to the stiff substrate. On the other hand, cells coming from the stiff side will stay along the boundary. These predictions guided the design of a durotaxis assay (2): Areas of different stiffnesses on which cells can freely move from soft to stiff and vice versa were created within the same micropillar substrate. Such substrates were composed of consecutive arrays of micropillars of diameters 1 and 2 µm (creating soft,  $k_s$ , and stiff,  $K_s$ , substrates, respectively) inducing a stiffness ratio of around ten between both sides  $(K_s/k_s \approx 10)$  while the surface density of the micropillars was kept constant (Fig. 5A). Cells coming from the soft part or from the stiff part were considered independently. As shown in Fig. 5D and in Movie S2 for  $k_s = 3$  nN/ $\mu$ m and  $K_s = 34$  nN/µm, cells indeed move toward the stiff part of the substrate aligning perpendicularly to the boundary.

We then used this assay to analyze the migration paths of individual cells on micropillar substrates of different stiffnesses. Our durotaxis analysis was performed by counting the preferential orientation of the cells after they touched the border between the two different parts of the substrate. We measured the number of events that corresponded to cells coming from the soft substrates and arriving at the boundary ( $40 < n_{cells} < 80$ , where  $n_{\text{cells}}$  is the total number of cells analyzed). By changing the rigidity of the soft micropillars from 3-134 nN/µm, we obtained an optimal durotaxis efficiency for stiffness values around 3-10 nN/µm with around 70% of cells moving toward the stiffer side (Fig. 5B). For  $k \gg 10$  nN/µm, no preferential direction was observed (only 35-50% of cells migrate toward the stiffer substrate). On the contrary, most of the cells located on the stiffer side did not migrate toward the soft one within the tested range of stiffnesses represented by the rigidity of the stiff pillars (Histogram, Fig. 5C, Movie S3). In addition, as a control experiment to test whether cells were actually reacting to the substrate rigidity and not the adhesion geometry, we fabricated adhesive surfaces with the same geometry on glass coverslips coated with fluorescently labeled fibronectin (SI Text, Fig. S2). To limit cell adhesion to the transferred pattern, the substrates were then treated with polyethyleneglycol to passivate the rest of the surface (26), rendering it nonstick. In these conditions, we obtained an array of fibronectin patches printed on glass with the same adhesive surface and sizes as the tops of the pillars. On such surfaces presenting the same geometry but no rigidity dependency, we observed that geometry could not explain our previous results on pillar substrates. Indeed, only 37% of the cells migrated from the area of 1  $\mu$ m patches to the area of 2  $\mu$ m ones and 53% in the opposite direction when first placed on the 2 µm patches area.

When both sides of the substrate presented a stiffness larger than around 50 nN/ $\mu$ m, we observed that the migration of cells toward the stiffer part was reduced, as if cells could not sense any difference between sides (Fig. 5B). In other words, an optimal response to a step difference in matrix stiffness appeared within a narrow range of rigidities. These results could be related to the ability of cells to adapt their own stiffness to that of the substrate (27, 43, 46). Moreover, the range of stiffness that promoted migration toward the stiffer side was correlated to that of the initial regime of the stress-stiffness relationship (that of fast stress increase before reaching saturation, see Fig. 3B). By analyzing crossing events with the deformation of micropillars at the boundary between soft and stiff substrates (Fig. 5), we showed that as a cell coming from the soft part was probing the interface, the force quickly increased up to 30 nN (Fig. 5F) and peaked in the direction n normal to the border, leading to the migration of the cell body toward the stiffest region (Fig. 5E, Movie S4). In agreement with our previous observations, this behavior can be attributed to cell polarization along this axis, which in turn favored the migration from soft to stiff.

It appears that migration toward the stiffer part was enhanced when the step in stiffness corresponded to a large increase in stress between both sides of the substrate. According to Fig. 3B, the range of rigidities between 1 and 50 nN/µm corresponds to the regime of the highest cell sensitivity to substrate rigidity. However, if both sides of the substrate corresponded to similar values of stress exerted by the adherent cell, cells did not migrate from one side to the other, no matter what the rigidity step between them. Consequently, the stress-stiffness relationship that we established appeared as an interesting indicator for the durotactic behavior. As cells sense the stiff side, they exert larger forces which in turn could induce a polarization of the actin cytoskeleton toward the stiff side and thus promote durotaxis. Interestingly, qualitatively similar behavior has been observed for the durotaxis of vascular cells on defined stiffness gradients within a range of rigidities up to 80 kPa (47). The study observed a higher polarized behavior on gradient gels than on uniform substrates and a durotactic behavior enhanced by increasing the magnitude of gradients. However, they did detect an upper limit of stiffness that a cell is capable of sensing. Here we show that such a limit could be determined by the relative variations of the stress that the cell could exert on both sides.



Fig. 5. Durotaxis revisited with micropillar substrates. (A) Scanning electron micrograph of a cell spread between two micropillar surfaces exhibiting different stiffnesses. The diameter of the pillars vary between 1 µm (top region) and 2 µm (bottom region), but the surface density of the micropillar substrate (fraction of surface are covered by pillars) has been kept constant. (Scale bar, 10 µm.) (B) Statistics of cells migrating from the soft substrate toward the stiff one as a function of the spring constant of the soft pillars. The spring constant, k, of the pillar was varied between 3 and 134 nN/µm. A high percentage of migration events toward the stiff substrate was observed for values between 3 and 11 nN/µm, in correlation with the range of stiffnesses where a high increase of the stress is observed. For larger values of k, no preferential direction was observed. In red, percentage of cells that come from the soft part and cross the boundary. In blue, percentage of cells that do not migrate. In gray, cells with undefined movement. (C) Statistics of cells migrating from the stiff part of the substrate as a function of the spring constant of the stiff pillars. Cells prefer to stay on the stiffer substrate over a wide-range of rigidity. In red, percentage of cells that come from the stiff part and prefer to stay on it instead of crossing the boundary. In blue, percentage of cells that migrate toward the soft substrate. In gray, cells with undefined movement. (D) Brightfield images of the movement of REF52 cell at the border between stiff ( $K_s = 34 \text{ nN}/\mu\text{m}$ ) and soft ( $k_s = 3 \text{ nN}/\mu\text{m}$ ) substrates. The dashed line represents the cell boundary. Time = 0 is here taken arbitrarily. (Scale bar, 20 µm.) (E) Force distribution during a typical crossing event of an REF52 cell from soft to stiff substrate. Micropillars are labeled by Cy3-fibronectin (red) and the cell appears in green. (Scale bar, 5 µm.) Cell spread at the boundary between soft and stiff substrates (1-µm pillars versus 2-µm pillars) exhibits a roughly homogeneous distribution of the forces on the soft part  $(k_s = 3 \text{ nN}/\mu\text{m} \text{ and } K_s = 34 \text{ nN}/\mu\text{m})$ . As the cell is probing a stiff pillar, a large deformation (i.e., high force) is peaked in the direction n, normal to the border inducing a polarized shape of the cell in this direction. (F) Variation of the force on the stiff pillar (designated by an asterisk in E, i) as a function of time.

#### Conclusion

We present here strong evidence that the rigidity-sensing mechanism in cell migration is not only locally driven by focal adhesion growth: The same size of FA area can lead to different forces depending on the substrate stiffness. Our results indicate clear constraints on the possible mechanisms that regulate cell mechanosensitivity. A nonconstant stress as a function of stiffness requires a large-scale mechanical feedback that involves the reorganization of actin stress fibers instead of a purely local mechanism based on FA mechanosensitivity. In our model, stress fibers may act as a force sensor that transmits a mechanical tension to FAs. Tension mediated through the actin cytoskeleton induces a polarization of actin fibers that orient along the direction of the applied force in response to substrate stiffness. Signaling pathways will help to regulate this tension by stabilizing FAs (48). We also demonstrate that durotaxis is optimal within a given range of rigidities that can be explained by our stress-rigidity relationship. The integrative

mechanism we have presented here could be relevant to understanding the matrix stiffness dependence of stem cell differentiation (3, 43) and also tumor formation in vivo (49) in terms of actin reorganization and cell contractility.

#### **Materials and Methods**

Cell Culture. REF52 cells were maintained at 37 °C in a humidified atmosphere of 5% CO<sub>2</sub> and 95% air in DMEM containing 10% bovine calf serum, 100 U/mL penicillin, 100 µg/mL streptomycin, and 100 µg/mL glutamine. The cells were deposited on µFSA 3 to 6 h before microscopy experiments (SI Text for details).

Preparation and Calibration of Microstructured PDMS Substrates. PDMS micropillar arrays were prepared and calibrated according to du Roure et al. (28). The SI Text provides additional details on the experimental methods.

Image Analysis. To calculate the order parameter, the images were analyzed using ImageJ software. First the images were convolved and thresholded.

Then we used the Analyze Particle tool to detect the actin filaments. Each filament was fitted with an ellipse. We then calculated the medium orientation of the filaments and the order parameter along this angle  $S = \langle \cos[2(\theta \cdot \langle \theta \rangle)] \rangle$ , where the average weighting was given by the length of the filaments (*SI Text*).

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## **Supporting Information**

### Trichet et al. 10.1073/pnas.1117810109

#### SI Text

**Cell Culture.** The cells were deposited on  $\mu$ FSA 3 to 6 h before microscopy experiments. Cells on micropillar substrates were fixed for 10 min at room temperature using 3% formaldehyde, 4% sucrose in PBS, rinsed in PBS, and permeabilized for 5 min with 0.5% Triton X-100 in PBS, after which they were blocked for 1 h with 3% BSA in PBS and rinsed again in PBS. The actin cytoskeleton was stained with Alexa 633 phalloidin at 1/1,000 dilution.

Preparation and Calibration of Microstructured Polydimethylsiloxane

(PDMS) Substrates. We used dimensionally calibrated macroscopic cylinders of this material and measured their compression under a fixed normal strain to determine the Young's modulus, E, of the PDMS. As E depends on the PDMS cure time, we used a consistent cure time of  $15 \text{ h} \pm 2 \text{ h}$  at 65 °C. E strongly depends on the fraction of the crosslinker added, which we varied from 3.3–10% corresponding to values ranging from E = 0.3 to 1.8 MPa. Above 5% crosslinker, we obtained a linear relationship between the fraction of crosslinker and the value of E. By performing scanning electron microscopy (SEM) observations, we measured the dimensions of the pillars. We used pillars with a diameter of 2 µm and a center-to-center distance of 4 µm. For the durotaxis experiment, the substrates were composed of consecutive arrays of micropillars of diameters 1 and 2 µm while the surface density of the micropillars was kept constant.

Only the top of the pillars was coated with fluorescently labeled fibronectin (Cy3; Amersham Biosciences). To achieve this coating, a stamp of flat PDMS was inked with a PBS solution containing 50  $\mu$ g/mL of fibronectin and 5  $\mu$ g/mL of labeled Cy3 fibronectin. The stamps were then dried and placed against the microforce sensor array ( $\mu$ FSA) for 10 min. Then we treated the substrates with pluronics (F127) for 1 h and rinsed with PBS 1× to prevent cell adhesion along the pillars.

**Time-Lapse Video-Microscopy.** Time-lapse images of cells on pillars were acquired using an Olympus BX51 upright microscope, equipped with an incubator maintaining the temperature at 37 °C.

For the durotaxis experiments, time-lapse sequences were acquired over 16 h using a frame delay of 5 min on the inverted microscope. Images were taken using an Olympus 20× air objective lens (N.A. 0.40). Using an automated stage, driven by Metamorph (Marzhauser), we observed different areas during the same experiment ( $40 < n_{cells} < 80$ ).

For the force measurements, time-lapse sequences were acquired over 30 min to 1 h using a frame delay of 30 s to 3 min on the upright microscope. The cells were observed with an Olympus  $60\times$  water immersion objective lens (N.A. 0.9). Pillars were imaged with the aid of the signal from the fluorescent fibronectin coating their tops. Both images (pillars and focal adhesions) were taken at each time point using the automated filter turret control, driven by Metamorph.

Image Analysis, Focal Adhesion Characterization, and Force Traction Measurements. We analyzed the time-lapse images using ImageJ software (National Institutes of Health). We determined the position of the adhesions for each picture and substracted the background. We then used the smooth filter, which replaced each pixel with the average of its  $3 \times 3$  neighborhood. All focal adhesions were thresholded and their areas over time were determined using the Particle Tracker plugin.

We used a homemade multiparticle tracking routine to detect micropillar positions for each image from the fluorescent signal emitted by the labeled fibronectin coating the top of the pillars. The contrast was high enough to determine the pillar position with a resolution of approximately 20–30 nm. We then calculated the displacement of each pillar by comparing the positions with the undeflected positions. The applied forces were deduced by multiplying the displacements by the spring constant of the pillars.

To relate the FA area and the force, we measured the corresponding FA area and pillar displacement. All data averaged measurements done on at least 15 different focal adhesions (or pillars) in at least five different cells.

Statistics collected on the focal adhesion area distribution involved analysis of 70 different focal adhesions in at least six different cells, placed on  $\mu$ FSAs of different rigidities.

Durotaxis analysis was performed by counting the preferential orientation of cells after touching the boundary between the two different parts of the substrate. Cells coming from the soft part or from the stiff part were considered independently. Each statistic refers to the analysis of 40 to 80 different cells. Statistics on microcontact-printing experiments were collected on 27 cells coming from the side with 2- $\mu$ m fibronectin-patches and 29 cells coming from the side with 1- $\mu$ m patches, respectively.

Scanning Electron Microscopy. Cells were cultured on a  $\mu$ FSA after their previous resuspension, fixed with 4% paraformaldehyde in PBS for 1 h and then further rinsed three times with PBS buffer. Dehydration was performed by rinsing the samples through graded ethanol/water mixtures (50, 70, 80, 90, and finally 100%, 10 min per step). Ethanol was slowly exchanged successively by amyl acetate and liquid CO<sub>2</sub>. Finally, samples were dried using the critical point method and then coated by sputtering with a thin layer of gold.

Theoretical Modeling of Cell Mechanosensitivity. Focal adhesions (FAs) play an important role in mechanotransduction, as shown by the reported dependence of their size on the sustained force. The usual generic mechanism invoked to account for this mechanosensitivity is that a force applied on an FA induces an elastic deformation of the contact that triggers conformational and organizational changes (such as unfolding) of some of its constitutive proteins, which in turn can enhance binding with new proteins and enable the growth of the contact. Initially, we give theoretical arguments based on a simple phenomenological model inspired by Nicolas et al. (1), which shows that FAs can act as mechanosensors and display an adaptive response to the local mechanical properties (elastic modulus) of their immediate environment, namely the extracellular matrix (ECM). We next show that this mechanosensitivity, based on mechanisms at the molecular scale, is only local, and cannot therefore explain the response to large-scale mechanical properties, such as the rigidity of pillar, as reported here. This failure of the simple model supports the existence of another mechanosensing mechanism, which we suggest could be mediated by the deformation of cytoskeletal structures such as stress fibers. Here our purpose is to provide theoretical arguments to support the existence of such a large-scale mechanosensing mechanism rather than to design an explicit model with specific functional forms.

*Local mechanosensitivity.* We first consider an FA bound to a flat substrate. We model the substrate and its coating of ECM molecules (such as fibronectin) as a linear elastomer of Young's mod-

ulus  $E_s$ . The FA plaque, of area  $S_{FA}$  and thickness d, is modeled as a homogeneous linear elastomer of Young's modulus  $E_{FA}$  and Poisson ratio  $\nu_{FA}$ . A longitudinal and homogeneous stress  $\sigma = F/S_{FA}$  is applied along the upper surface of the FA (direction x). We denote by  $u_s$  the deformation field (constant in a first approximation) along x at the substrate/FA interface, and  $u_{FA}$  the deformation field of the upper surface of the FA. Given the timescales involved (mins) in the mechanical response of the cell, we assume the forces are balanced and obtain, as follows:

$$F = \sigma A = g_{\rm FA} u_{\rm FA} S_{\rm FA} = k_s u_s, \qquad [S1]$$

where  $g_{\text{FA}} = E_{\text{FA}} / [2d(1 + \nu_{\text{FA}})]$  and  $k_s \propto E_s$ . Following Nicolas et al. (1, 2), we first assume that the dynamics of  $S_{\text{FA}}$  is coupled to the deformation of the substrate  $u_s$ . This coupling can be written generically as

$$\frac{dS_{\rm FA}}{dt} = f(u_s, S_{\rm FA}),$$
[S2]

where f is an increasing function of  $u_s$  which typically decreases for large  $S_{\text{FA}}$ , defining a steady state area  $S_{\text{FA}ss}(u_s)$  by  $f(u_s, S_{\text{FA}ss}) = 0$ .

This phenomenological model accounts for the effect mentioned above that an elastic deformation of the substrate induces conformational changes of ECM molecules and makes new binding sites for FA proteins available. Typically, a linear dependence for small  $u_s$  can be expected. The explicit form of f depends on the microscopic modelling and will not be discussed here [Nicolas et al. (1)] for a possible functional form. In particular, this model shows that the kinetics of FA growth depends on the applied force (because  $u_s = F/k_s$ ), as observed in experiments (3).

For the FA to function as an autonomous mechanosensor, the stress  $\sigma$  applied by the stress fiber on the FA has to be regulated by the FA itself, even if indirectly. This feedback can be generically written as follows:

$$\sigma = h[u_{\rm FA}], \qquad [S3]$$

where the brackets means functional dependence. This constitutive equation determines the stress applied by the stress fiber thereby completing the set of equations, Eqs. **S1–S3**. As above, a hypothetical molecular basis of this feedback relies on the shear-induced conformational changes of FA proteins, which enhance actin polymerization on the FA plaque and therefore increase the stress. Note finally that refinements of this model can be added, for instance to account for shear stiffening of the substrate, by taking  $g_{FA}(u_s, u_{FA})$ .

The point that we make here is that the Eqs. **S1–S3** fully determine the unknowns  $S_{FA}$ ,  $\sigma$ ,  $u_s$ ,  $u_{FA}$  as functions of  $E_s$  only as parameter of the environment of the FA. This set of equations defines very generally the local mechanosensitive response of the FA, already observed in previous studies (3). Here our purpose is to provide a generic outline; we therefore do not give explicit functional forms.

Large-scale mechanosensitivity. We now show that the mechanosensitivity effect that we report in this paper is not due only to the previously reported mechanism mediated by FAs, and that it requires a larger scale mechanism. We keep the same notations as in previous paragraph, except that now the substrate is not infinite, but elastically bound to a wall by a spring of stiffness k, accounting for the pillar elasticity. Note that k can be varied by tuning the geometric parameters of the pillars and keeping  $E_s$ constant.

We denote by x the displacement of the pillar along the force direction. All the equations in the previous paragraph still hold (in the shifted frame of the pillar surface, and assuming that the local deformation  $u_s$  is much smaller than the global displace-

ment x), and mechanical equilibrium gives the extra equation F = kx for the extra variable x. This equation is independent of Eqs. **S1–S3** above, which shows straightforwardly that if the feedback is mediated by the FA only as above, then  $S_{\text{FA}}$ ,  $\sigma$ ,  $u_s$ ,  $u_{\text{FA}}$  will depend on  $E_s$  only and not on k, which can be varied independently of  $E_s$  through the geometric parameters of the pillar. However, our experimental findings are that the stress depends on k (Fig. 3A). We therefore conclude that extra feedback involving other structures than FAs must be involved. We suggest below that this feedback could be mediated by the cytoskeleton, and in particular by stress fibers pulling on the FAs. We therefore write instead of Eq. **S3**:

$$\sigma = h[x(t)].$$
 [S4]

The system (Eq. S1, S2, S4) is now fully coupled, and generically the resulting stress will now depend on k, as observed experimentally. As a result, a large-scale mechanosensing mechanism realized by the coupling [S4] must exist.

**Cytoskeleton-based mechanosensitivity.** The coupling (Eq. S4) means that a deformation of the cytoskeleton (here a shortening of length x of a stress fiber) impacts on the active stress exerted by the cytoskeleton. On general grounds, such coupling can be justified in the framework of the theory of viscoelastic active gels (4). This theory states by symmetry arguments that in polar or nematic media driven out of equilibrium such as the cytoskeleton, there exists an active stress  $\sigma_{ij}^a$  (here due to actin/myosin interactions) proportional to the nematic tensor  $Q_{ij} = \langle n_i n_j \rangle - \delta_{ij}/3$  and reads  $\sigma_{ij}^a = \alpha Q_{ij}$  where  $\alpha$  is a phenomenological coupling constant. Here  $n_i$  denotes the orientation vector of a single actin filament and the average is taken locally over a coarse-graining length scale. In the case of a quasi two-dimensional adhering cell, we obtain

$$Q_{\theta_0\theta_0} = \langle \cos 2(\theta - \theta_0) \rangle, \qquad [S5]$$

which quantifies the polarization along the direction  $\theta_0$ , where  $\theta$ denotes the angle of n with a reference axis. The scalar order parameter S is then defined as  $S = Q_{\langle \theta \rangle \langle \theta \rangle}$ , where  $\langle \theta \rangle$  is the preferred direction of the system. Because imaging of single actin filaments is not accessible in our set-up, stress fibers can be used as a coarse-grained order parameter to evaluate  $Q_{ii}$ . In generic visco elastic nematic materials, the nematic tensor is in turn coupled to the strain  $U_{ii}$ . More qualitatively, this coupling means that a local deformation x induces a strain  $U_{ii}$  in the cytoskeleton and therefore a reorganization of the orientation  $Q_{ii}$  of the filaments, which in turn yields an active stress. This coupling between  $Q_{ii}$  and  $U_{ii}$  can be verified experimentally. We found that a strong dependency of the order parameter with the rigidity of the substrate. In the case of rigid substrate ( $k = 100 \text{ nN}/\mu\text{m}$ ) (small strain  $U_{ii}$ ), the cells are polarized so that  $\langle S \rangle \approx 0.8$  (Fig. 4C). On the contrary, in the case of a soft substrate ( $k = 4 \text{ nN}/\mu\text{m}$ ) (large strain  $U_{ii}$ ), stress fibers showed no preferred direction, yielding  $S \approx 0.19$  (Fig. 4A). At intermediate stiffness ( $\approx 40 \text{ nN}/\mu\text{m}$ ), we obtained  $\langle S \rangle = 0.56$  (Fig. 4*B*).

In a first approximation, as in the model introduced in Zemel et al. (5), a linear coupling can be suggested of the form,

$$\sigma^a = -\gamma (U - U^0), \qquad [S6]$$

where  $U^0$  is the initial contractile strain and  $\gamma$  is a fourth rank tensor (indexes have been dropped) which determines the active response of the cell to deformation. Such coupling was found compatible with observations of the response of stem cells to matrix stiffness in Zemel et al. (5).

Such a model can be made more explicit at the scale of a single stress fiber, which is a one-dimensional structure with well-defined boundary conditions, because stress fibers end at focal adhesions. In this case Eq. **S6** can be rewritten as

$$\sigma^a = -\gamma (x - x^0).$$
 [S7]

The force balance condition then yields  $F = kx = \sigma^a S_{FA}$  where  $S_{FA}$  denotes the area of the focal adhesion ending the stress fiber. We finally obtain

$$\sigma^a = \frac{k\gamma x^0}{k + \gamma S_{\rm FA}},$$
 [S8]

which can be fitted to experimental results of Fig. 3B. Additionally, we can obtain the force which reads, as follows:

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$$F = \frac{k\gamma x^0 S_{\rm FA}}{k + \gamma S_{\rm FA}}.$$
 [S9]

This functional form of the force as a function of substrate stiffness is compatible with the model introduced in Zemel et al. (5), which is also based on linear response. In the regime of small stiffness k, it yields to linear order  $F \approx kx^0$ , which corresponds to the experimental results of Fig. 2C. The parameter  $x^0$ , which can be fitted from the data to the value  $x^0 \approx 800$  nm, can be interpreted as the typical shortening length of a stress fiber. For a typical stress fiber length of 10 µm, this result is compatible with observations (6) where a 5–15% shortening was obtained.

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Fig. S1. Distribution of focal adhesions areas on micropillar substrates and flat PDMS substrates.

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Fig. S2. Micropatterned substrates for durotaxis control experiment. Cell adhesion on glass micropatterned substrates at the boundary between 1- and 2-µm fibronectin patches (red). The density of fibronectin patches was kept constant between both sides of the substrates. Actin cytoskeleton is labeled in green.



**Movie S1.** An example of the traction forces over time exerted by an REF52 cell expressing YFP-paxillin (green) on micropillars stamped with fluorescent fibronectin (red). (Scale bar, 10 μm.) The duration of the movie is 2 h. Movie S1 (AVI)



**Movie S2.** An example of the cell movements at the boundary between a stiff substrate ( $K_s = 80 \text{ nN}/\mu\text{m}$ ; pillar diameter = 2  $\mu$ m) and a soft substrate ( $k_s = 7 \text{ nN}/\mu\text{m}$ ; pillar diameter = 1  $\mu$ m). The cell first comes from the soft part of the substrate and transmigrates toward the stiff part with a perpendicular orientation with regards to the boundary.

Movie S2 (MOV)

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**Movie S3.** An example of the cell movements at the boundary between a stiff substrate ( $K = 80 \text{ nN}/\mu\text{m}$ ; pillar diameter = 2  $\mu$ m) and a soft substrate ( $k = 7 \text{ nN}/\mu\text{m}$ ; pillar diameter = 1  $\mu$ m). The cell is coming from the stiff part of the substrate, probes the soft part, rotates, but stays on the stiff side. Movie S3 (MOV)

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Movie 54. Local deformations of the substrate at the interface between soft and stiff parts. Pillars are in red and cell staining in green. The transmigration process toward the stiff part induces a sudden increase of the force normal to the boundary. Movie 54 (MOV)

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